Supporting Information for

## Monolayer Graphitic Carbon Nitride as Metal-Free Catalyst with Enhanced Performance in Photo- and Electro-Catalysis

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# **Supplementary Tables and Figures**

Table S1 Summary of tri-s-triazine-based g- $C_3N_4$  nanosheets and monolayer. Synthetic methods and characterization are listed

Precursor	Method	Thickness	Characterization (Morphology)	Characterization (XRD)	Compare with bulk g-C <sub>3</sub> N <sub>4</sub>		References
					FT-IR	XPS	
Dicyanamide	Bottom up	0.4-0.5	None	Crystallinity	Same	Same	[S1]
Dicyanamide	Bottom up	0.4	TEM (nanosheet image)	Crystallinity	Same	Same	[S2]
Melamine	Bottom up	0.38	SEM, TEM (nanosheet image)	Crystallinity	Same	Same	[S3]

Melamine	Bottom up	0.5	SEM, TEM (nanosheet image)	Crystallinity	Same	Same	[S4]
Melamine	Bottom up	0.64	SEM, HRTEM (nanosheet image)	Crystallinity	Same	Same	[\$5]
Urea	Bottom up	0.6	TEM (nanosheet image)	Crystallinity	Same	Same	[S6]
Melamine	Bottom up	0.32	SEM, TEM (nanosheet image)	Crystallinity	Same	Same	[S7]
Urea	Bottom up	0.34	SEM, TEM (nanosheet image)	Crystallinity	Same	Same	[S8]
Cyanamide	Top down	0.45	Cs-HRTEM (atomic image)	Amorphous	Blue shift	Blue shift	This work



Fig. S1 Tyndall effect and zeta potential of the g-C<sub>3</sub>N<sub>4</sub>-m colloidal suspension.



**Fig. S2** Typical high resolution-transmission electron microscopy (HR-TEM) images of  $g-C_3N_4-m$ . **a-c** Transmission electron microscopy (TEM) images of randomly chosen  $g-C_3N_4-m$ . **d-f** A magnified view of the area marked by the yellow box in c-e, respectively. The  $g-C_3N_4-m$  m exhibited well-known orderly structure with hexagonal symmetry along with triangular cavity formed by encirclement of three adjacent heptazine units on the surface



**Fig. S3** Unit cell structures with the simulated through-focal series spherical aberration (Cs) corrected high-resolution transmission electron microscopy (Cs-HRTEM) images for  $g-C_3N_{4-}$  b. **a** Top view of super unit cell and side view of unit cell. **b** The simulated through-focal series showing the effect of objective lens defocus on image contrast and the corresponding atomic identification. The simulation program operated was MacTempas. (the acceleration voltage of 80 kV, Cs of 0.034 mm, the spread of defocus of 30 nm, the beam convergence of 2 mrad, the specimen thickness of 1 nm and the defocus value of a step of 6 nm in the range from 54 to 300 nm were used for the simulation. **c, d** Magnified images of the areas marked by the yellow (f = 90 nm) and red boxes (f = 0 nm) in **b** 



**Fig. S4** Unit cell structures with the simulated through-focal series Cs-HRTEM images for g- $C_3N_4$ -m. **a** Top view of super unit cell and side view of unit cell. **b** The through-focal series simulation showing the effect of objective lens defocus on contrast in images and the corresponding atomic identification. The simulation program was MacTempas. The acceleration voltage of 80 kV, Cs of 0.034 mm, the spread of defocus of 30 nm, the beam convergence of 2 mrad, the specimen thickness 1 nm and the defocus value of a step of 6 nm in the range from 54 to 300 nm were used for the simulation. **c, d** Magnified images of the areas marked by the yellow (f = 276 nm) and red (f = 48 nm) boxes in b, respectively

In general, atomically thin and freely suspended 2D supramolecular entities, such as the monolayers of  $g-C_3N_4$ -m in solution as shown in this study, are not expected to be perfectly flat; i.e. they are rather flexible and probably exhibit a microscopic corrugation or a randomly curved single molecular sheet. Such a microscopic crumpling has often raised a problem in detailed HRTEM analysis, since the phase contrast seems to be extremely sensitive particularly for 2D compounds to defocus (*f*), spherical aberration (Cs) and alignment conditions of the

objective lens, which means that there are often multiple defocus values at different microscopic corrugating positions, even within the same HRTEM image. In this study, the defocus series of simulated images were performed to assist in interpreting experimental Cs-HRTEM images. Under a certain condition, for instance, with 276 nm defocus, the white dots are coinciding with the atom positions as shown in Fig. S4c, but in the case with 48 nm defocus, a reversed contrast can be seen as well demonstrated in Fig. S4d. After performing detailed simulation studies, we were able to find a systematically periodic structure with a triangle arrangement of bright white dots and a triangular cavity in the hexagonal, which is well consistent with the structure model shown in Fig. S4a.



**Fig. S5** Selected area electron diffraction (SAED) patterns of  $g-C_3N_4$ -b. **a** A low-magnification TEM image of  $g-C_3N_4$ -b. **b** Cs-HRTEM image which is magnified from the yellow box in a. **c** The experimental diffraction pattern and **d** the simulated one. The electron beam is given along the [0001] g-C\_3N\_4-b axis



**Fig. S6** Cs-HRTEM images of  $g-C_3N_4$ -b. **a** Low-magnification TEM image of  $g-C_3N_4$ -b. **b** The image magnified from the yellow box in a. **c** A Wiener-filtered Cs-HRTEM image experimentally observed, which is magnified from the yellow box in b. And the inset is theoretically simulated image with the corresponding atomic model overlaid for  $g-C_3N_4$ -b



**Fig. S7** Annular dark field scanning transmission electron microscopy (ADF-STEM) images of **a**  $g-C_3N_4$ -b and **b**  $g-C_3N_4$ -m were marked with red color-coded lines for the corresponding electron energy loss spectroscopy (EELS) data

Element	Atomic % Theoretical	g-C <sub>3</sub> N <sub>4</sub> –b Experimental	g-C <sub>3</sub> N <sub>4</sub> –m Experimental	
С	42.9	44.2	43.2	
Ν	57.1	55.8	56.8	
N/C Ratio	1.33	1.26	1.32	



Fig. S8 X-ray photoelectron spectra (XPS) of a C1s and b N1s for g-C<sub>3</sub>N<sub>4</sub>-b and g-C<sub>3</sub>N<sub>4</sub>-m

The C 1s peak here for the bulk could be deconvoluted into three peaks at 284.8, 286.3 and 288.1 eV, corresponding to the signals from the standard reference carbon, the carbon atoms in C-NH<sub>2</sub> and the sp<sup>2</sup>-hybridized carbon in the aromatic ring (N-C=N), respectively [S9]. The binding energy for the N1s spectrum was also deconvoluted into three peaks, 398.7, 400.0 and 400.9 eV, corresponding to the sp<sup>2</sup>-hybridzed aromatic N bonded to carbon atoms (C-N=C), tertiary N bonded to carbon atoms in the form of N-(C<sub>3</sub>), and to the quaternary N bonded to three carbon atoms in the aromatic rings (C-NH<sub>2</sub>), respectively [S10]. An additional small peak at 404.5 eV was ascribed to the  $\pi$ - $\pi$  excitation occurring between layers [S3].



**Fig. S9** Atomic force microscope (AFM) image of  $g-C_3N_4$ -n. **a** The  $g-C_3N_4$ -n dispersed on the freshly-cleaved muscovite mica substrate. **b** The height profile along the white dashed line in a



Fig. S10 Photocatalytic activity for visible light-induced H<sub>2</sub>O<sub>2</sub> generation

The excellent photocatalytic property of g-C<sub>3</sub>N<sub>4</sub>-m was also evidenced by the visible light ( $\lambda$ > 420 nm)-induced generation test of H<sub>2</sub>O<sub>2</sub>, and the photocatalytic formation rate of H<sub>2</sub>O<sub>2</sub> on g-C<sub>3</sub>N<sub>4</sub>-m was determined to be twenty times and three times faster than that on g-C<sub>3</sub>N<sub>4</sub>-b and g-C<sub>3</sub>N<sub>4</sub>-n



**Fig. S11 a** Linear sweep voltammograms (LSV) curves of ORR, **b** Tafel plots, and Nyquist plots measured at **c** OCV and **d** 0.6 V

The monolayer  $g-C_3N_4$ -m showed excellent electro-catalytic properties compared to the bulk and the nanosheets, due to the fact that the formation of monolayers resulted in a remarkable increase in active sites.

### **Supplementary References**

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